1	Perovskite oxygen carrier with chemical memory under reversible
2	chemical looping conditions with and without SO <sub>2</sub> during reduction
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11	Abstract
12	Oxygen carrier materials (OCM) are usually exposed to sulfur-contained gases in the fuel
13	reactor for chemical looping combustion. This work provides both experimental and model work to
14	understand the SO <sub>2</sub> effect on the heterogeneous redox kinetics of a $CaMn_{0.375}Ti_{0.5}Fe_{0.125}O_{3-\delta}$ -based
15	perovskite oxygen carrier. The cycle reactivity and redox kinetics under reducing conditions were
16	conducted with and without SO <sub>2</sub> in a micro-fluidized bed thermogravimetric analysis technology
17	(MFB-TGA). The redox kinetic behaviors were simulated by a bubbling fluidized bed reactor model
18	coupled with a two-stage kinetic model. The SO <sub>2</sub> can react with the perovskite to increase the
19	oxygen transfer capacity from 4 wt% to 5 wt%. When the temperature is higher than 1173 K, $SO_2$
20	has almost no effect on the $H_2$ reduction reactivity, while the oxidation reactivity decreases by 50 %,

22 lower than 1173 K, there is a significant sulfur-poisoning effect during oxidation and reduction. The

but the oxidation is still fast enough to achieve 4 wt% capacity within 8 s. When the temperature is

analyses of XRD, SEM-EDS, and in-situ DRIFTS indicated that most of the absorbed sulfur mainly existed in the sulfate/sulfide shell on the particle surface. The chemical kinetics and physical structure of  $CaMn_{0.375}Ti_{0.5}Fe_{0.125}O_{3-\delta}$  perovskite can be completely recovered in the absence of SO<sub>2</sub>, and this perovskite oxygen carrier is chemically memorable and reversible in its solid structure. The fundamental understanding of the sulfur effect on the redox kinetics and solid structure of the perovskite oxygen carrier provides a new insight to the material development and corresponding reaction mechanisms.

30 Keywords: Chemical looping combustion; Oxygen carrier material; CaMnO<sub>3</sub>, Perovskite oxide;
 31 sulfur; Kinetics

## 32 **1. Introduction**

Chemical looping combustion (CLC) has emerged as a promising technology to capture CO2 33 34 from fossil fuel combustion for power and heat generation [1,2]. A typical CLC unit usually consists 35 of an air reactor and a fuel reactor. A solid oxygen carrier material (OCM) circulates between the reactors, to deliver the required lattice oxygen for fuel conversion in the fuel reactor. Then, the 36 37 reduced OCM experiences an oxidization step in the air reactor. Developing an OCM with fast 38 reactivity, high stability, as well as high attrition resistance, is the key to the CLC technology [3]. 39 Up to now, both natural minerals and manufactured OCMs have been applied in the CLC process 40 [3,4]. Generally, these OCMs are metal oxides from manganese (Mn) [5,6], iron (Fe) [7], copper 41 (Cu) [8], nickel (Ni) [9], etc. To further improve the physical and chemical properties of OCMs, 42 some bimetallic OCMs have been developed, such as Cu-Fe-based [10], Ni-Mn-based [11], and Cu-43 Mn-based [12] OCMs, etc. For transition metal oxide OCMs, phase change during redox cycles has 44 often been observed, resulting in materials degradation, agglomeration, i.e. shorter life time.

45	Recently, synthetic OCMs with perovskite-type structures (ABO <sub>3-<math>\delta</math></sub> ) have attracted increasing
46	interest for CLC applications thanks to the fast oxygen bulk diffusion and surface exchange during
47	the consecutive redox cycles, as well as their redox and stability which can be tuned by varying
48	doping elements and their concentrations on both A and B sites [13-15]. Enormous efforts have been
49	devoted to improving OCMs' reactivity and recyclability performance of OCMs based on CaMnO3-
50	$_{\delta}$ . For example, researchers substituted Ti, Fe or Mg into CaMnO <sub>3-<math>\delta} structure to selectively change</math></sub>
51	the cationic site, to improve reactivity, stability, crushing strength, fluidization property, etc. [14,
52	16-21]. The investigations mainly focused on the determination of chemical redox reactivity and
53	heterogeneous gas-solid reaction kinetics at variable temperatures and gas atmospheres (O2, H2, CO,
54	CH <sub>4</sub> ), and partial pressures under sulfur (S)-absence condition [17-19]. The CaMn <sub>x</sub> B <sub>1-x</sub> O <sub>3-<math>\delta</math></sub> (B=Fe,
55	Ti, Mg etc.) OCMs have shown promising chemical properties for the CLC process as long as sulfur
56	(S) is absent [17-19, 22]. However, sulfur (S) is widely present in fossil fuels, which means the
57	OCM has to expose to sulfur-containing species in the fuel reactor. The previous studies found that
58	the chemical reactivity of the $CaMn_xB_{1-x}O_{3-\delta}$ suffered from deactivation due to calcium sulfate
59	formation, once exposed to the situation of sulfurous fuels [23-25]. In these researches, CH <sub>4</sub> with
60	different sulfur contents was used as the fuel to assess the chemical performance of OCMs [23-25].
61	However, the major reducing gas compositions for solid fuels are CO and $\mathrm{H}_2$ from the
62	devolatilization and gasification steps in the fuel reactor of CLC, and CH4 content is normally very
63	low in the fuel reactor. Therefore, the conclusions or results regarding sulfur effect on chemical
64	performance by utilizing CH4 fuel, cannot be used directly when evaluating solid fuel CLC, e.g.
65	reactor design. Up to now, the determination of heterogeneous reaction kinetics and insight to the
66	influence mechanism of the fuel in the presence of SO2 are very limited due to insufficient

experiments and characterizations. In the previous work,  $CaMn_{0.375}Ti_{0.5}Fe_{0.125}O_{3-\delta}$  particles as OCMs were controllably fabricated via spray drying granulation, and the heterogeneous reaction kinetics was investigated without SO<sub>2</sub> containing gases [26]. Heterogeneous redox reaction kinetics in the presence of SO<sub>2</sub> is more closely associated with the chemical performance evaluation, reactor design and optimization for the solid fuel CLC. However, the influence of SO<sub>2</sub> on the reactivity and redox kinetic behaviors is still partially unknown.

73 In this work, the investigation of oxidization and reduction kinetics with and without SO<sub>2</sub> was 74 conducted by means of a micro-fluidized bed thermogravimetric analysis technology (MFB-TGA), 75 based on real-time mass change measurement of OCM particles in a fluidizing state close to 76 enhanced mass and heat transfer in a real CLC reactor [27,28]. The redox kinetic behaviors were 77 described by a simplified K-L model coupled with a two-stage kinetic model via changing the 78 chemical reaction rate constant ( $k_{chem}$ ,  $k_{diff}$ ), and the ratio of the conversion level at the fast reaction stage to the conversion level ( $\Psi$ ). The OCM was characterized by X-ray Fluorescence (XRF), 79 80 scanning electron microscope (SEM), energy dispersive spectrometer (EDS) mappings, X-ray 81 diffraction (XRD), and in-situ reflectance infrared spectroscopy (DRIFTS).

82 **2. Experimental** 

### 83 2.1. Oxygen carrier material

The perovskite oxide,  $CaMn_{0.375}Ti_{0.5}Fe_{0.125}O_{3-\delta}$ , was fabricated as the OCM particles employing spray granulation followed by a calcination stage at 1623 K for 12 h [26]. Particle size in the range of 180-250 µm was used to investigate the reaction kinetics. X-ray Fluorescence (XRF) results show that the compounds of the oxidized OCM are 40.52 wt% CaO, 26.38 wt% TiO<sub>2</sub>, 22.94 wt% MnO and 7.07 wt% Fe<sub>2</sub>O<sub>3</sub>. More properties of the OCM are listed in Table 1. The true density (ρ<sub>t</sub>) was measured by a helium replacement method (AccuPyc II 1345, Micromeritics). The porosity
was analyzed by a mercury intrusion porosimetry (AutoPore IV 9500, Micromeritics). The surface
area was obtained by Brunauer–Emmett–Teller (BET) using a nitrogen adsorption instrument at 77
K (ASAP 2460, Micromeritics).

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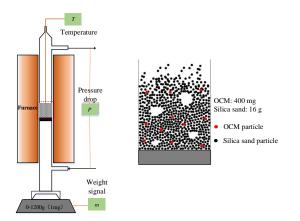
Table 1 Physical properties of the oxidized and reduced CaMn<sub>0.375</sub>Ti<sub>0.5</sub>Fe<sub>0.125</sub>O<sub>3-δ</sub>

State	$d_{\rm p}(\mu{\rm m})$	$\rho_{t}(kg/m^{3})$	Porosity	BET(m <sup>2</sup> /g)
Oxidized	180-250	4910	0.493	0.26
Reduced	180-250	4898	0.489	0.35

94

#### 95 **2.2.** Micro-fluidized bed thermogravimetric analysis (MFB-TGA)

96 The chemically oxidizing and reducing kinetics of the perovskite particles was determined 97 via a micro-fluidized bed thermogravimetric analysis technology (MFB-TGA) [27,28]. The main 98 components include a heating furnace, a fluidized-bed reactor, a gas supply unit, and a measurement 99 unit, among which, the core is the fluidized-bed reactor with an inner diameter of 30 mm (Fig. 1). 100 The flowrates of the fluidizing agents were controlled at 1.2-1.5 L/min (STP), which ensures that 101 the fluidized-bed reactor was operated at a bubbling state during the measurements. The switch 102 among different fluidizing agents into the bottom of the bubbling-bed reactor uses well-connected 103 magnetic valves. A mass transducer with a readability of 1 mg was employed to measure the on-line 104 mass change caused by chemical gas-solid reactions. A K-type thermocouple enclosed in the 105 bubbling-bed reactor was employed to measure bed temperature. A differential pressure transducer 106 was employed to monitor the pressure drop of the bubbling-bed reactor. More details can be found 107 elsewhere [26-28].



108 109

Fig. 1. Schematic diagrams of the MFB-TGA apparatus

For a typical experiment of redox kinetics,  $\sim 16$  g silica sand with the particle size at 300-355 110 111  $\mu$ m was firstly filled as the inert bed material, followed by ~ 400 mg fully oxidized OCM particles 112 injected into the bubbling-bed reactor after heating up to the desired temperature under a 21 vol% 113 O<sub>2</sub> fluidizing agent (Fig. 1). After the stable mass and temperature signals were reached, fluidizing 114 gases switch between the reducing gases (10 % H<sub>2</sub> with 500 ppm SO<sub>2</sub> or 10 % H<sub>2</sub> without SO<sub>2</sub>) and 115 oxidizing gas (21 vol% O<sub>2</sub>) was conducted for redox cycles. Inert gas was flowed into the reactor for 30 s between the oxidizing and reducing steps, to prevent direct mixing of  $H_2$  and  $O_2$ . 500 ppm 116 117  $SO_2$  was mixed with the reducing gas to create a sulfur exposure atmosphere and simulate a more 118 real condition in the CLC process of the sulfur-containing fuels. Operating at the superficial velocity 119 of 3-5  $U_{\rm mf}$ , the oxidization and reduction rates were obtained at 1023-1173 K. Redox cycles were 120 repeated at least 3 times to ensure reproducibility at each temperature. Table 2 concludes the 121 experimental scheme of the kinetic determination under condition with or without SO<sub>2</sub>. Series I-IV 122 were used to investigate the evolution of reaction kinetics under reduction condition with SO<sub>2</sub>. 123 Series V was performed to regenerate the sulfur-poisoned OCM under the reduction condition 124 without SO<sub>2</sub>. Finally, the redox kinetics of the poisoned and recovered OCM at 1023-1173 K was 125 determined. Another multicyclic test, similar to Series I-IV, was performed under reducing condition

#### 126 without $SO_2$ for comparison.

Series	Temperature	Fluidizing agents	Periods	No of cycles
Ι	1173 K	O <sub>2</sub> /H <sub>2</sub> (SO <sub>2</sub> )	120s/480s	2
II	1173 K	O <sub>2</sub> /H <sub>2</sub> (SO <sub>2</sub> )	120s/180s	3
III	1173K	O <sub>2</sub> /H <sub>2</sub> (SO <sub>2</sub> )	60s/180s	2
IV	1173K	O <sub>2</sub> /H <sub>2</sub> (SO <sub>2</sub> )	60s/120s	43
V	1173K	O <sub>2</sub> /H <sub>2</sub> (no SO <sub>2</sub> )	60s/120s	15

127 Table 2. Experimental scheme of determining the reaction kinetics under sulfur-presence/absence

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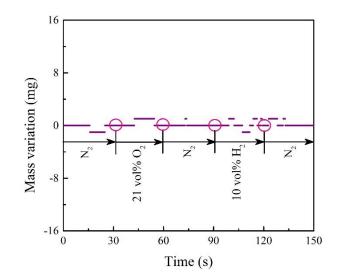
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The balance gases in the reaction agents (10 vol% H<sub>2</sub> with 500 ppm SO<sub>2</sub> or 10 vol% H<sub>2</sub> without SO<sub>2</sub> and 21 vol% O<sub>2</sub>) were well matched to ensure similar densities as inert gas (33.22 vol% He in 66.78 vol% Ar), see Table 3. In this regard, the noise possibly created by gases switch was eliminated. Finally, the impact of the gas switches on the on-line mass signals was eliminated, and the measurement error of the MFB-TGA is kept within  $\pm 1$  mg, see Fig. 2. Therefore, the on-line mass change was only caused by the heterogeneous gas-solid reactions, as shown from a typical redox cycle in Fig. 3.

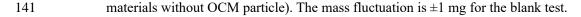
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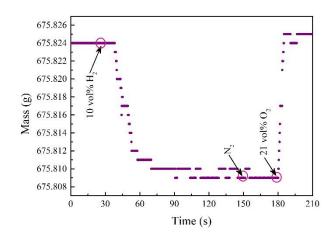
Table 3. The compositions of the reaction gases and the balance gases

Reaction gas	Inert gas
10 vol% H <sub>2</sub> +21.65 vol% He + 68.35 vol% Ar	33.22 vol% He+66.78 vol% Ar
10 vol% H <sub>2</sub> +500 ppm SO <sub>2</sub> +21.65 vol% He + 68.35 vol% Ar	33.22 vol% He+66.78 vol% Ar
21 vol% O <sub>2</sub> +28.59 vol% He + 50.41 vol% Ar	33.22 vol% He+66.78 vol% Ar



140 Fig. 2. Blank test of the gas switch at 1173 K (only 16.4 g silica sand was filled as the bed





142

143 Fig. 3. A typical redox test at 1173 K: the bed materials consists of 16 g silica sand and 400 mg

144 OCM particles, and the experimental data were obtained by switching the reaction agents between

- 145 10 vol% H<sub>2</sub> and 21 vol% O<sub>2</sub>.
- 146 **2.3. Post characterizations**

147 To determine possible sulfur effects on the surface OCM particles, in-situ diffuse reflectance

- 148 infrared Fourier transform spectroscopy (NEXUS, 870 FT-IR) was used. The mixture of 80 wt%
- 149 KBr and 20 wt% OCM powders were added into a photo-reactor. A background IR spectrum was

collected in pure Ar (500 mL/min) atmosphere after pretreated at 993 K. After Ar purging for an additional 30 min, the reduction by SO<sub>2</sub>-containing H<sub>2</sub> (10 vol% H<sub>2</sub> with 500 ppm SO<sub>2</sub>) was conducted at 993 K for 480 s. After reduction, pure Ar was used to flush the reactor for 120 s. Then, 21 vol% O<sub>2</sub> was introduced for 300 s in the oxidization step, followed by another Ar purging step for 120 s. Subsequently, the above reduction and oxidization steps were repeated for 48 cycles. The IR spectra were obtained in real-time during the test.

The phase purities were obtained by X-ray diffraction (XRD) patterns (D8 Advance, BRUKER). Scans were selected between 10° and 90° with a step of 10°/min to analyze the phase compositions. The morphology characteristics of the surface and cross-section of the tested CaMn<sub>0.5</sub>Ti<sub>0.375</sub>Fe<sub>0.125</sub>O<sub>3-δ</sub> particles were characterized by a scanning electron microscope (SEM, Zeiss Merlin), and the element distributions were analyzed by an energy dispersive spectrometer (EDS).

#### 162 **2.4. Data evaluation**

163 The on-line mass signals, m(t), were recorded by the mass transducer during the test. The 164 oxygen transfer capacity ( $R_{oc,t}$ ), defined as Eq. (1), was experimentally estimated as explained 165 below. The conversion levels for the reduction ( $X_r$ ) and oxidization ( $X_o$ ) reactions were calculated 166 as Eqs. (2) and (3), respectively.  $m_{oc}$  is the mass of the added OCM particles in the fully oxidized 167 state, while  $m_o$  and  $m_r$  are the total mass of the reactor and bed materials, corresponding the OCM 168 particles in fully oxidized and reduced states, respectively.

169 
$$R_{o,t} = \frac{m_o - m_r}{m_{oc}}$$
 (1)

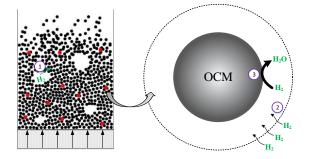
170 
$$X_{\rm r}(t) = \frac{m_{\rm o} - m(t)}{R_{\rm ot} m_{\rm oc}}$$
 (2)

171 
$$X_{o}(t) = \frac{m(t) - m_{r}}{R_{ot}m_{oc}}$$
 (3)

### 172 **3. Models**

180

In the bubble bed reactor, the OCM particles were averagely distributed in the emulsion phase together with the silica sand, whereas the reaction gases mainly existed in the bubble phase. Taking the reduction processes as an example (Fig. 4), the reaction gases in the fluidizing agents are firstly diffused from the bubble phase to the emulsion phase. After the interphase gas exchange, the reaction gases are further transferred to the OCM particle surface distributed in the emulsion phase. Finally, the gas-solid reaction occurs on the OCM particles. Therefore, there is a requirement to apply a reactor model to analyze the interphase gas diffusion in the reactor.



1) Interphase gas interchange; 2) External gas diffusion; 3) gas-solid reaction

Fig. 4. Schematic diagrams of the reduction reaction with H<sub>2</sub> in the bubbling bed reactor To describe the mass transfer and chemical reaction processes in the bubbling bed reactor, a simplified reactor model was applied here, and expressed by Eqs. (4) and (5) for the bubble phase and emulsion phase according to mass balance [28]. The reactor model was based on the classic

185 fluidization bed reactor model put forward by Kunii and Levenspiel (K-L model) [29].

186 
$$-U_b^* \frac{\mathrm{d}C_{\mathrm{b},i}}{\mathrm{d}z} = K_{\mathrm{be}}(C_{\mathrm{b},i} - C_{\mathrm{e},i}), (i = \mathrm{O}_2, \mathrm{H}_2)$$
(4)

187 
$$-(1-\theta)U_{\rm mf}\frac{\mathrm{d}C_{\rm e,i}}{\mathrm{d}z} = -\theta K_{\rm be}(C_{\rm b,i} - C_{\rm e,i}) + f_{\rm a}(1-\theta)(1-\varepsilon_{\rm mf})K_{\rm r}C_{\rm e,i}, (i=\mathrm{O}_2,\mathrm{H}_2)$$
(5)

188 where  $C_{b,i}$  (mol/m<sup>3</sup>) and  $C_{e,i}$  (mol/m<sup>3</sup>) is the gas concentration in the bubble phase and emulsion 189 phase, respectively.  $U_b^*$  (m/s) and  $U_{mf}$  (m/s) refer to the effective gas velocity in the bubble 190 phase and the minimum fluidization velocity.  $K_{be}$  (s<sup>-1</sup>) is the gas interchange coefficient between 191 the bubble phase and the emulsion phase.  $f_a$  is the volume fraction of the OCM particles in all 192 solid volume.  $\theta$  is the fraction of the bubble phase, and  $\varepsilon_{mf}$  is the voidage of fluidized bed at 193 minimum fluidization state. The supporting document introduces how to calculate the above 194 coefficients and the analytical solution of Eqs. (4) and (5) in detail.  $K_r$  (s<sup>-1</sup>) is the comprehensive 195 reaction rate constant, including the gas-solid reactions and the impact of the external mass transfer 196 resistance, expressed as [18,26]:

<sup>197</sup> 
$$K_{\rm r} = \frac{1}{\frac{1}{K_{\rm ri}} + \frac{d_{\rm p}}{6k_{\rm g}}}$$
 (6)

198 
$$K_{\rm ri} = \frac{R_{\rm o,t}\rho_{\rm s}}{\alpha M_{\rm o}} \left\{ k_{\rm chem} (\psi X_{\rm e} - X_{\rm chem})^{2/3} + k_{\rm diff} \left[ (1 - \psi) X_{\rm e} - X_{\rm diff} \right]^{2/3} \right\}$$
(7)

199 
$$k_{\rm g} = (Sh \cdot D_i)/d_{\rm p}, \ (i = O_2, H_2)$$
 (8)

where  $k_{g}$  (m/s) is the external mass transfer coefficient,  $D_{i}$  (m<sup>2</sup>/s) is the molecular diffusivity of 200 gas *i*,  $d_{\rm p}$  (m) is the particle size of the used oxygen carrier, Sh is Sherwood number.  $K_{\rm ri}$  (s<sup>-1</sup>) 201 is the reaction rate constant, describing the gas-solid reactions only, and the value of  $K_{\rm ri}~({\rm s}^{-1})$  is 202 203 dependent on the conversion level of the oxygen carrier particles. A kinetic model was coupled with 204 the reactor model here. It is well known that a gas-solid reaction commonly includes two stages, i.e., an initially fast stage and a second slow stage [30]. Some kinetic models have been developed for 205 206 this purpose from literature, such as the apparent model [31], the shrinking core model [32], the pore model [33], and the grain model [34]. However, the above models fail to predict the kinetic 207 208 transition behavior from the initially fast reaction stage to the second slow reaction stage [30]. In 209 this work, a semi-empirical model containing the two reaction stages was applied to describe the 210 conversion of the oxidation and reduction reactions, as seen in Eqs. (9)-(11) [18,26].

211 
$$\frac{dX_{chem}}{dt} = k_{chem} (\psi X_e - X_{chem})^{2/3} C_{e,i}, \quad (i = O_2, H_2)$$
(9)

212 
$$\frac{\mathrm{d}X_{\mathrm{diff}}}{\mathrm{d}t} = k_{\mathrm{diff}} [(1 - \psi)X_{\mathrm{e}} - X_{\mathrm{diff}}]^{2/3} C_{\mathrm{e},i}, \quad (i = \mathrm{O}_2, \mathrm{H}_2)$$
(10)

213 
$$X = X_{\text{chem}} + X_{\text{diff}}$$
(11)

214 where  $X_{\rm chem}$  and  $X_{\rm diff}$  are the conversion levels of the first stage and the second stage,  $X_{\rm e}$  is 215 the equilibrium conversion level. The second stage starts at the end of the fast reaction stage (critical conversion,  $\psi X_e$ ).  $k_{chem}$  (m<sup>3</sup>/mol/s) and  $k_{diff}$  (m<sup>3</sup>/mol/s) represent the reaction rate constants of 216 217 the fast reaction stage and the slow reaction stage, respectively.  $\Psi$  refers to the kinetic transition 218 point. The exponent of the two stages, 2/3, is a known parameter in the model. Finally, the total conversion level of a reaction is the sum of the two stages. Based on the kinetic model, a conversion 219 versus time curve can be obtained by giving the values of the rate constant ( $k_{chem}$  (m<sup>3</sup>/mol/s) and 220  $k_{\text{diff}}$  (m<sup>3</sup>/mol/s)) and ratio ( $\psi$ ). Then the value of  $K_{\text{ri}}$  (s<sup>-1</sup>) can be calculated by using Eq. (4). 221 222 As a result, the kinetic model is coupled with the reactor model through  $K_r$  (s<sup>-1</sup>).

In the model, the oxidization by O<sub>2</sub> and reduction by H<sub>2</sub> are assumed to be first-order [17,19,22]. It should be noted that the reactor model implemented with the two-stage kinetic model has an analytical solution, and the detailed calculation algorithm can be found elsewhere [18,26]. In the model,  $k_{chem}$  (m<sup>3</sup>/mol/s),  $k_{diff}$  (m<sup>3</sup>/mol/s) and  $\psi$  depend on the reaction temperature, which

227 can be expressed as Arrhenius equations,

228 
$$k_{\rm chem} = A_{\rm chem} \exp\left(-\frac{E_{\rm chem}}{R_g T}\right)$$
 (12)

229 
$$k_{\text{diff}} = A_{\text{diff}} \exp\left(-\frac{E_{\text{diff}}}{R_g T}\right)$$
 (13)

230 
$$\psi = A_{\psi} \exp\left(-\frac{E_{\psi}}{R_g T}\right)$$
 (14)

231

The axial distribution of the gas concentration is variable in the emulsion phase. In this work,

the average gas concentration across the whole bed in the emulsion phase calculated by Eq. (15)
was applied to calculate the conversion levels in Eqs. (9) and (10).

234 
$$C_{e,i,ave} = z^{-1} \int_0^z C_{e,i} dz, \ (i = O_2, H_2)$$
 (15)

## 235 4. Results and discussion

#### 236 4.1. Redox multicycles

237 The redox multicycles of the OCM at 1173 K are shown in Fig. 5. In Fig. 5(a), the whole redox process was done under the reduction condition without SO<sub>2</sub>. The mass variation profiles are quite 238 239 stable during the multicycles, implying there is no chemical degradation under the reduction 240 condition without SO<sub>2</sub>. The total oxygen transfer capacity is kept at approximately 4 wt% during 241 the redox multicycles. In Fig. 5(b), the 1-50 redox cycles were conducted under the reducing 242 condition with SO<sub>2</sub>. It is clear that the mass variation profiles tend to rise. The mass variation profiles 243 indicate the total oxygen transfer capacity of the OCM is progressively increasing with the redox cycle number under the reduction condition with SO<sub>2</sub> until 24 redox cycles, see Fig. 5(b). After that, 244 245 the mass variation profiles are stable for the following 20 redox cycles, implying a chemical stability 246 state under the reduction condition with SO<sub>2</sub>. Afterwards, the reducing gas was altered to 10 vol% 247 H<sub>2</sub> without SO<sub>2</sub>. On the contrary, the mass variation profiles stay progressively decreasing from 248 cycle 51 to cycle 62, following a new chemical stability state under the reduction condition without 249  $SO_2$ . From the fluctuation of the pressure drop profiles (Figs. 5(a) and (b)) and the high-temperature 250 fluidization (Video 1), there is no observed sintering and agglomeration during the redox tests. It 251 can be concluded that the OCM particles possess perfect fluidization properties.

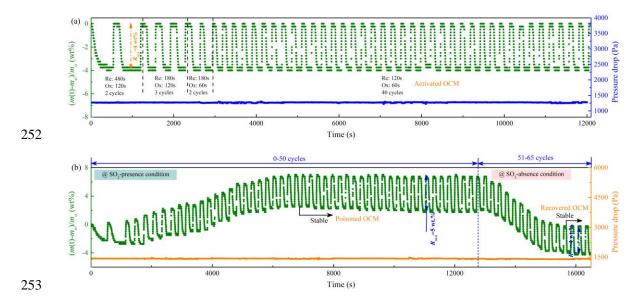
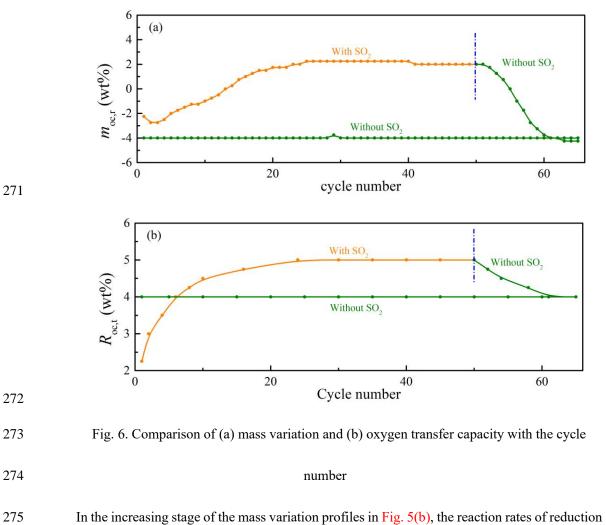


Fig. 5. Multicycles of mass variation and pressure drop during redox cycles at 1173 K
(oxidization: 21 vol% O<sub>2</sub>; reduction: 10 vol% H<sub>2</sub>). (a) without SO<sub>2</sub>; (b) with SO<sub>2</sub>, reduction (1-50
cycles) : 10 vol% H<sub>2</sub> with 500 ppm SO<sub>2</sub>; reduction (51-65 cycles) : 10 vol% H<sub>2</sub> without SO<sub>2</sub>.
Fig. 6 compares the mass variation and oxygen transfer capacity during the redox multicycles.
In Fig. 6(a), *m*<sub>oc,r</sub> refers to the mass change after reduction of each cycle, and the change of *m*<sub>oc,r</sub>, the
with the cycle number stands for the "baseline" change during the redox multicycles. At *m*<sub>oc,r</sub>, the
OCM is under the reduced state. From the profiles in Fig. 6(a), the value of *m*<sub>oc,r</sub> experiences an

200	
261	increase from -2.25 wt% to 2 wt% with the redox cycle number when reduced with $SO_2$ -containing
262	H <sub>2</sub> . After that, the value of $m_{oc,r}$ is kept unchanged from cycle 24 to cycle 50. Once switching the
263	reduction condition to SO <sub>2</sub> -free H <sub>2</sub> , the value of $m_{oc,r}$ is decreasing to -4.25 wt% as the redox
264	proceeds. Compared with the redox process without SO <sub>2</sub> , the value of $m_{oc,r}$ stays at -4 wt% from the
265	beginning to the end. The presence of $SO_2$ in the reduction step also affects the oxygen transfer
266	capacity $(R_{oc,t})$ of the OCM, as seen in Fig. 6(b). If the OCM is reduced without SO <sub>2</sub> , the value of
267	$R_{oc,t}$ is always equal to 4 wt% during the whole redox multicycles. However, reduction in the
268	presence of SO <sub>2</sub> would change the oxygen transfer capacity. In the whole redox process, the total

oxygen transfer capacity increases continuously from 2.3 wt% to 5 wt% under the reduction



### 270 condition with $SO_2$ , then decreases to 4 wt% after removing $SO_2$ .

275 In the increasing stage of the mass variation profiles in Fig. 5(b), the reaction rates of reduction 276 are enhanced from cycle 1 to cycle 24 (Fig. 7(a)). There is no noticeable change in reduction rates 277 between cycle 24 and cycle 50. The depth of reduction increases from 2.25 wt% to 5 wt%, indicating 278 that more lattice oxygen can be utilized during the reduction step. When reduced without SO<sub>2</sub>, the 279 reduction rates of the OCM are slightly improved with the redox cycle number, but the depth of 280 reduction finally stabilizes to 4 wt%, see Fig. 7(b).

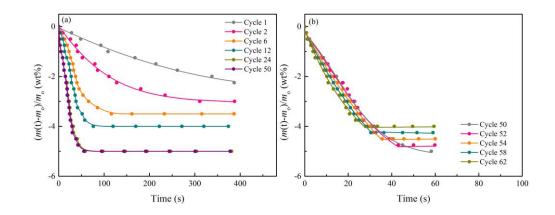
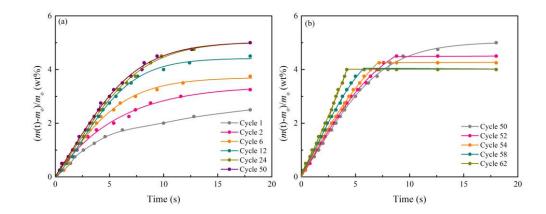


Fig. 7. Mass variation with the cycle number for the reduction process in Fig. 5(b). (a) with SO<sub>2</sub> and (b) without SO<sub>2</sub>

284 During the first 24 cycles in Fig. 5(b), the regeneration of oxidization can take up more oxygen 285 (O) elements in the OCM, see Fig. 8(a). For example, the mass loss is 4 wt% for the reduction of cycle 12, but 4.25 wt% oxygen (O) elements are taken up during the oxidization step. The gained 286 oxygen (O) element by CaS oxidation to CaSO<sub>4</sub> can play an active role for the following reduction 287 288 step. Hence, the oxygen transfer capacity is rising for the former 24 redox cycles. On the contrary, from cycle 50- 60 in Fig. 5(b), fewer oxygen (O) elements are taken up during the regeneration of 289 oxidization steps due to removal of active CaS-CaSO<sub>4</sub> shell type OCM. The oxygen transfer 290 291 capacity finally decreases to 4 wt%.



293

281

Fig.8. Mass variation with the cycle number for the oxidization process in Fig. 5(b). (a) with



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During the multiple redox cycles under the reduction condition without SO<sub>2</sub>, the change of chemical performance with the cycle number is demonstrated in Figs. 9(a) and (b). The reduction rate is continuously improved as the cycle number increases (Fig. 9(a)), but the oxidization rate can stabilize since cycle 6 (Fig. 9(b)). Hence the OCM shows an activation process during the redox multicycles. The reduction depth always stays at 4 wt%, hence the total oxygen transfer capacity stays unchanged during the whole process.

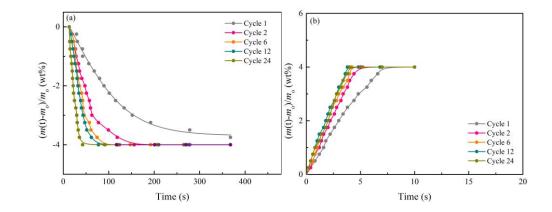


Fig. 9. Mass variation with the cycle number for the (a) reduction process and (b) oxidization
process under reduction condition without SO<sub>2</sub> in Fig. 5(a)

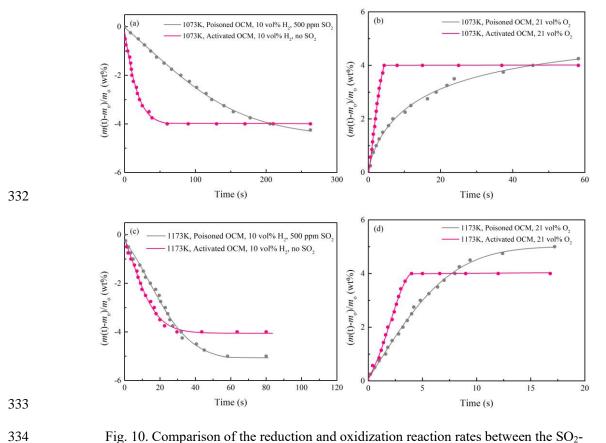
301

It should be noted here that the perovskite oxide after 24 redox cycles is termed as "Activated OCM", as illustrated in Fig. 5(a). The "Activated OCM" is not treated by SO<sub>2</sub>. The OCM in the chemical stability state under the reduction condition with SO<sub>2</sub> is called "Poisoned OCM", whereas the "Recovered OCM" means the chemically stable OCM particles under the reduction condition without SO<sub>2</sub>, as depicted in Fig. 5(b). It can be seen that the total oxygen transfer capacity of the "Activated OCM" is equal to that of the "Recovered OCM".

### **4.2. Comparison of the oxidization and reduction reaction rates**

Figs. 10(a) -(d) compare the reduction and oxidization reaction rates for the SO<sub>2</sub>-poisoned and
 activated OCMs at 1073 K and 1173 K. When reacting at 1173 K, the reduction rate of the poisoned

313	OCM is very close to that of the activated OCM, and the required time for reduction to 4 wt% is
314	$\sim$ 28 s for the activated OCM and $\sim$ 35 s for the poisoned OCM, respectively. However, the poisoned
315	OCM can be further reduced due to a higher total oxygen transfer capacity. When the temperature
316	decreases to 1073 K, the reduction rate of the activated OCM just decays a little, and the required
317	time for full reduction is prolonged to $\sim 60$ s. Whereas the poisoned OCM is deeply affected by
318	temperature, and the reduction rate of the poisoned OCM decreases by $\sim$ 7 times, as compared to
319	that at 1173 K. One common point for both the poisoned OCM and the activated OCM is that the
320	reduction curves are composed of an initially fast reaction stage and a second slow reaction stage.
321	For the oxidization reaction, the reaction rate of the poisoned OCM is slower than that of the
322	activated OCM at 1073 K and 1173 K. For the activated OCM, the temperature has a weak impact
323	on the oxidization rate, and the oxidization is only dominated by the initial fast reaction stage.
324	However, for the poisoned OCM, the oxidization rate decreases a lot with decreasing temperature,
325	and the oxidization curves consist of a fast reaction stage and a slow reaction stage, similar to the
326	reduction curves. Although the oxidation kinetics decreased by 50% as compared to that of the
327	activated OCM 1173 K, the oxidation of the SO <sub>2</sub> -poisoned OCM is still fast enough and can achieve
328	4 wt% of oxygen transfer capacity within 8 s. Fig. 10 shows that the existence of $SO_2$ in the reducing
329	gas would cause a change in both the reduction and oxidization behaviors, although an increase in
330	the oxygen transfer capacity. The OCM shows decreased oxidization and reduction rates due to
331	sulfur poisoning, especially reaction at low temperatures.



poisoned and the activated OCM (the activated perovskite oxide is not treated by SO<sub>2</sub>) at 1073 K
and 1173 K. (a) reduction at 1073 K; (b) oxidization at 1073 K; (c) reduction at 1173 K; (d)

337 oxidization at 1173K

From Figs. 11(a) and (b), the OCM can automatically recover from the SO<sub>2</sub>-poisoning state once treating it with SO<sub>2</sub>-absence fuel. Not only can the total transfer capacity be recovered to the original state, but also the recovered OCM shows the nearly same oxidization and reduction rates as the Activated OCM. Hence, it is quite interesting to notice that the investigated OCM possesses a chemical memory.

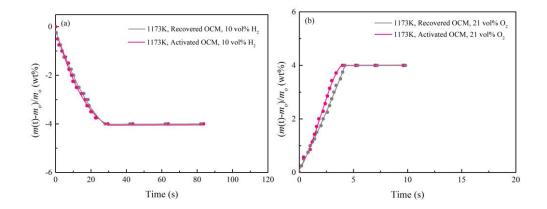


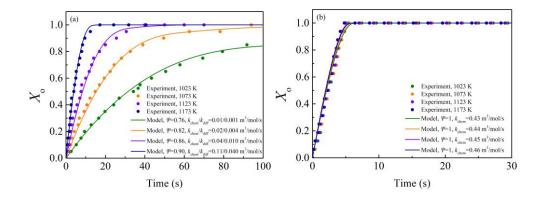
Fig. 11. Comparison of the (a) reduction and (b) oxidization reaction rates for the recovered and the activated OCM (the activated perovskite oxide is not treated by SO<sub>2</sub>) at 1173 K.

54.

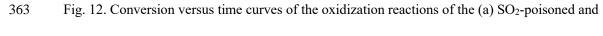
343

### 346 **4.3 Kinetic modelling results**

347 The model results and experimental data of the oxidization reactions of the SO<sub>2</sub>-poisoned and 348 recovered OCMs are shown in Figs. 12(a) and (b). The conversion levels at different temperatures 349 are plotted as a function of time. It is quite obvious that, the plots of conversion vs. time for the 350 SO<sub>2</sub>-poisoned OCM show two-stage behavior, i.e., an initially fast stage followed by a second slow stage. The model results agree well with experimental data. The reaction temperature has a 351 352 significant effect on the critical conversion ( $\psi X_{e}$ ) and reaction rates for the oxidization step. Again, 353 the calculated results could reproduce the experimental data obtained from the recovered OCM as 354 well (Fig. 17(b)). Since the whole oxidization process merely has a fast stage, the value of  $\Psi$  is set to 1 during the model calculation. The required time for the full oxidization is only ~4 s. The 355 356 oxidization rate is insensitive to the temperature. The above kinetic behaviors of the oxidization 357 reactions are similar to our previous work [26], where the reaction kinetics was driven by the activated OCM under the SO<sub>2</sub>-free fuel, which implies that the oxidization performance is 358 359 completely regenerated from the SO<sub>2</sub>-poisoned state. After recovering from the SO<sub>2</sub>-poisoning state, the chemical reaction rate constant  $(k_{\rm chem})$  for the oxidization reaction is higher than that of the 360



361 poisoned OCM, owing to the regeneration of the perovskite structure.



364 (b) recovered OCM at 1023-1173 K (oxidization: 21 vol%  $O_2$ )

Figs. 13(a) and (b) show the model and experimental results of the reduction reactions of the 365 SO<sub>2</sub>-poisoned and recovered OCM. The similarity between these two plots is that the reduction 366 process consists of an initially fast stage followed by a second slow stage, like the oxidization of the 367 368 SO<sub>2</sub>-poisoned OCM. The distinction is that the reduction rate of the SO<sub>2</sub>-poisoned OCM is 369 apparently slower than that of the recovered OCM. For example, the full reduction needs  $\sim 160$  s for 370 the SO<sub>2</sub>-poisoned OCM at 1123K, which is ~4 times slower than that of the recovered OCM. There 371 is a significant effect of the reaction temperature on the critical conversions  $(\psi X_{e})$  for the reduction 372 steps, and the reduction rate can be enhanced with increasing temperature. The kinetic behaviors of 373 the reduction reactions are also similar to the activated OCM [26], because of the full release of the absorbed sulfur in the perovskite structure. As a result, the chemical reaction rate constants (  $k_{
m chem}$ 374 375 and  $k_{\rm diff}$  ) for the reduction reaction are higher for the recovered OCM.

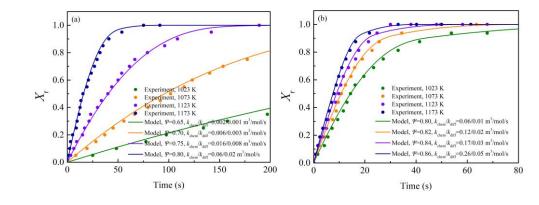


Fig. 13. Conversion versus time curves of the reduction reactions of the (a) SO<sub>2</sub>-poisoned and (b)
recovered OCM at 1023-1173 K (reduction: 10 vol% H<sub>2</sub>)

379 From the Arrhenius plots of  $\ln k_{chem}$ ,  $\ln k_{diff}$ , and  $\ln \psi$  (Figs. 14(a)-(c)), the activation energies 380 and pre-exponential factors can be obtained from the slopes and intercepts, respectively. The kinetic 381 parameters are summarized in Table 4. Influenced by sulfur, the activation energies of the oxidization and reduction reactions of the SO2-poisoned OCM are 156.64 kJ/mol and 222.56 kJ/mol 382 383 at the first reaction stage, which is far higher than that of the recovered OCM, because of more 384 sensitivity to the temperature. Table 4 also includes the kinetic parameters of the activated OCM obtained by similar reaction conditions [26]. Due to inherent chemical memory, the obtained kinetic 385 386 parameters for the recovered OCM are similar as those of the activated OCM. According to the 387 Arrhenius plots, the sulfur effect on the oxidization and reduction kinetics can be weakened with 388 increasing temperature. For example, the sulfur effect on the reduction reactivity could be removed when elevating the temperature to ~1323 K. The above kinetic results are reliable since the oxygen 389 transfer kinetics is faster than the oxygen consumption by the chemical reactions (Figs. S1-S4). 390

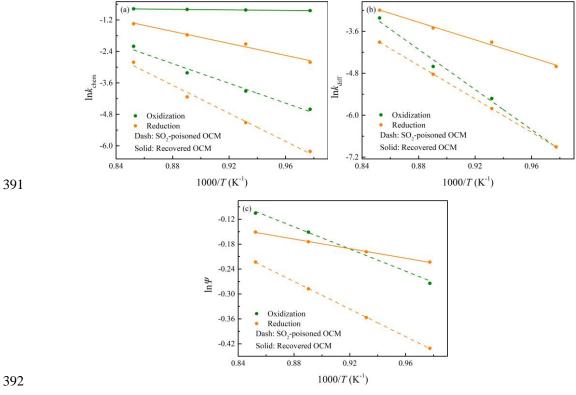




Fig. 14. Arrhenius plots of (a)  $\ln k_{\text{chem}}$ , (b)  $\ln k_{\text{diff}}$ , and (c)  $\ln \psi$ 

	SO <sub>2</sub> -poisoned OCM		Recov	Recovered OCM		Activated OCM <sup>[26]</sup>	
	O <sub>2</sub>	H <sub>2</sub>	O <sub>2</sub>	H <sub>2</sub>	O <sub>2</sub>	H <sub>2</sub>	
$A_{\rm chem} ({ m m}^3/{ m mol/s})$	9.05×10 <sup>5</sup>	4.26×10 <sup>8</sup>	0.92	4.25×10 <sup>3</sup>	0.92	$4.25 \times 10^{3}$	
$E_{ m chem}$ (kJ/mol)	156.64	222.56	6.69	94.17	6.69	94.17	
$A_{\rm diff}~({ m m}^3/{ m mol/s})$	1.56×10 <sup>9</sup>	1.45×10 <sup>7</sup>	n.a.	$1.62 \times 10^{3}$	n.a.	$1.62 \times 10^{3}$	
$E_{ m diff}$ (kJ/mol)	238.89	199.04	n.a.	100.73	n.a.	100.73	
$A_{\!$	2.82	3.26	n.a.	1.43	n.a.	1.43	
$E_{arphi}$	11.12	13.81	n.a.	5.04	n.a.	5.04	

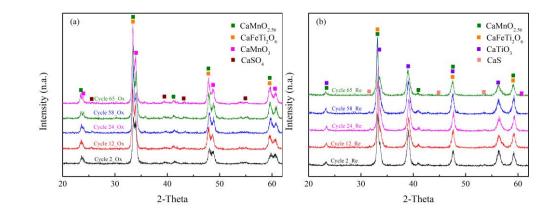
394 Table 4. Summary of the kinetic parameters of the SO<sub>2</sub>-poisoned, recovered, and activated OCM

# 396 **4.4. Sample characterization**

397

The fully oxidized and reduced particles extracted from variable redox cycle numbers in Fig.

5(b) were characterized by X-ray diffraction (XRD), see Figs. 15(a) and (b). The diffraction peaks
of CaSO<sub>4</sub> (PDF 37-1496) and CaS (PDF 08-0464) emerge under the reduction condition with SO<sub>2</sub>
but disappear under the reduction condition without SO<sub>2</sub>. The common crystal phases of the
oxidized and reduced OCM particles are CaMnO<sub>2.56</sub> (PDF 45-1267) and CaFeTi<sub>2</sub>O<sub>6</sub> (PDF 83-0265),
and the distinction is CaMnO<sub>3</sub> (PDF 03-0830) from the oxidized one and CaTiO<sub>3</sub> (PDF 082-0228)
from the reduced one.



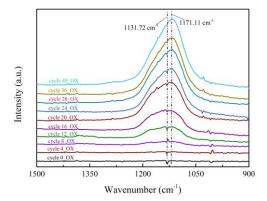
405 Fig. 15. X-ray diffraction patterns of the (a) oxidized and (b) reduced perovskite oxide at
406 different cycles (cycle 2, 12, 24, 58, 65) in Fig. 5(b)

404

407 In situ diffuse reflectance infrared Fourier transform spectroscopy (in-situ DRIFTS) also shows 408 that the CaSO<sub>4</sub> phase (1131.72 cm<sup>-1</sup> and 1171.11 cm<sup>-1</sup>) is gradually formed on the OCM particle 409 surface under the reduction condition with SO<sub>2</sub>, as shown in Fig. 16. Therefore, the gained mass in 410 Fig.5(b) is also contributed by the sulfate and sulfide. The added oxygen transfer capacity due to 411 formation of CaSO<sub>4</sub>/CaS on the surface of oxygen carrier brings the oxygen transfer capacity to 5 412 wt% during the redox cycles under the reduction condition with SO<sub>2</sub>. The profiles in Figs. 6(a) and 413 6(b) show sulfur effect leads to an increase in the oxygen transfer capacity from 4 wt% to 5 wt%, 414 while the mass increased by  $\sim 6 \text{ wt\%}$  (Fig. 6(a)). If this is CaO to CaS and all CaS formed is ideally active, the one should expect the formation of CaSO<sub>4</sub> to give an increase in the oxygen transfer 415

416 capacity of 24 wt%. Actually, only 1 wt% oxygen transfer capacity was obtained due to sulfur effect.

The possible explanation is the CaS to CaSO<sub>4</sub> oxidation has a low CaS conversion, and high percentage of the CaS reactant remains in the solid residues, according to Madarasz et al. [35] and Wang et al. [36].



420

421 Fig. 16. In situ diffuse reflectance infrared Fourier transform spectroscopy (In situ-DRIFTS)

### 422 during redox cycles

423 The scanning electron microscopy (SEM) images of the oxidized and reduced particles from 424 various redox cycle numbers indicate that the particle surface is gradually covered by a white shell 425 after reduction condition with exposure to  $SO_2$ , as depicted in Figs. 17(a) and (b). The grain 426 boundary on the particle gradually becomes invisible due to the formation and coverage of the white 427 shell. According to the energy dispersive spectrometer (EDS) results, the "white" sections on the 428 particles are the enrichment region of oxygen (O), sulfur (S), and calcium (Ca) for the oxidized 429 OCM and the enrichment region of sulfur (S) and calcium (Ca) for the reduced one, as seen in Figs. 430 S5 and S6. Therefore, the components of the white shells on the oxidized and reduced OCM particle 431 are identified as sulfate and sulfide respectively, which agrees well with the XRD patterns and in 432 situ-DRIFTS results. Note that the formed shell is not very dense at sulfur -saturated state. After 433 exposure to the reduction condition without SO<sub>2</sub>, the sulfate and sulfide shells will gradually 434 disappear, and the OCM particle can restore to the original state.

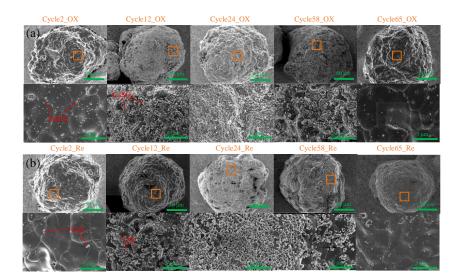
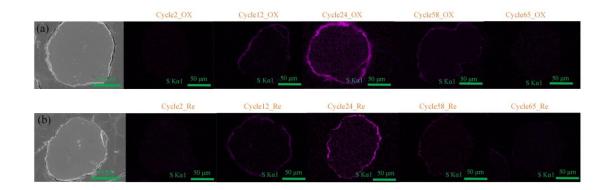


Fig. 17. Scanning electron microscopy images of the (a) oxidized and (b) reduced OCM
particles at different cycles (cycle 2, 12, 24, 58, 65) in Fig. 5(b)

438 To identify whether the bulk of the OCM particle is affected by the sulfur (S) element, energy 439 dispersive spectrometer (EDS) mapping of the particle cross-sections was further performed. With 440 the increasing redox cycle, sulfur (S) element was gradually accumulated on the particle surface as 441 a sulfate/sulfide shell, c.f. Figs. 18(a) and (b). Meanwhile, the average distribution of the sulfur (S) 442 elements was detected inside the OCM particle as well. The sulfur (S) element ratio of the surface 443 to the bulk is approximately 15:1 (Fig. S7), indicating that the sulfur (S) element is mainly enriched 444 on the particle surface. Such enrichment of sulfur (S) element on the shell shows that the OCM 445 particle possesses sulfur tolerance and self-protection ability. Nevertheless, a small amount of sulfur (0.06 wt%) still affected the lattice structure inside the OCM particle, possibly caused by SO<sub>2</sub> 446 447 molecules entering the particle inside through the pores or grain boundaries during the reduction 448 process.



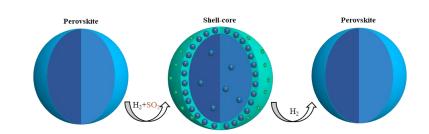
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450 Fig. 18. Sulfur (S) element distribution in the cross-sections of the (a) oxidized and (b)
451 reduced OCM particles at different cycles (cycle 2, 12, 24, 58, and 65) in Fig. 5

452

### 4.5 Possible Mechanism

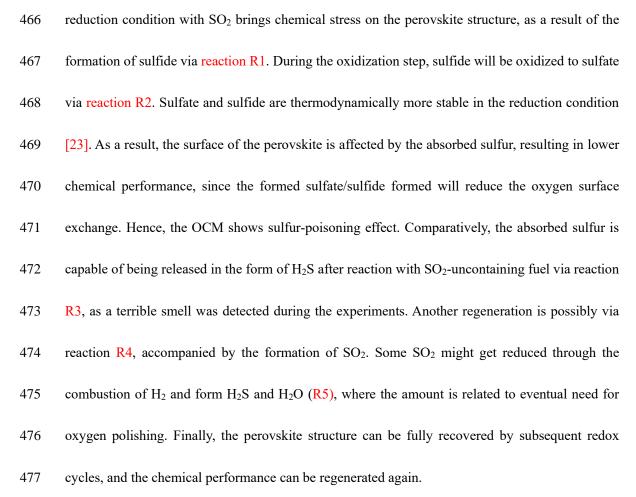
453 To conclude, the whole OCM particle is initially the perovskite oxide. When the OCM particle 454 is exposed to the reduction condition with SO<sub>2</sub>, the particle surface would form a non-dense sulfate/sulfide shell, which wraps the OCM particle. The core is weakly affected by sulfur. Hence a 455 shell-core structure forms under the reduction condition with SO<sub>2</sub>. The reaction gas molecules can 456 457 still contact the OCM particle through the non-dense shell. Once reduced in the reduction condition without SO<sub>2</sub>, almost all sulfur in the shell and core will be gradually released from the particle. 458 459 Finally, the shell-core structure disappears, and the whole OCM particle restores to its initial state, 460 as illustrated in Fig. 19.

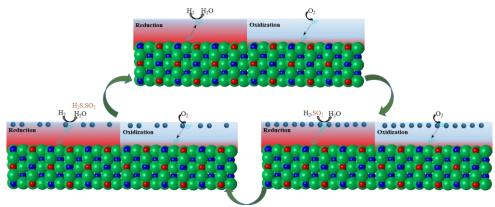


461 462

Fig. 19. The formation and disappearance of the non-dense sulfate/sulfide shell

463 The effect of sulfur on the chemical reactions can be diagramed in Fig. 20. On the reactant 464 surface of the activated OCM, the oxygen anions can be smoothly exchanged with  $H_2$  or  $O_2$  through 465 gas-solid reactions. Hence, both the reduction and oxidization rates are fast. Exposing to the





479 Fig.20. Schematic illustrations of the sulfur-poisoning and recovery process in the perovskite

### 480 structure (red: calcium; blue: manganese/iron/titanium; green: oxygen)

- 481  $CaO+3H_2+SO_2 \rightarrow CaS+3H_2O$  R1
- 482  $CaS+2O_2 \rightarrow CaSO_4$  R2
- 483  $CaSO_4 + 4H_2 \rightarrow CaO + H_2S + 3H_2O$  R3

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484 
$$CaSO_4 + H_2 \rightarrow CaO + SO_2 + H_2O$$
 R4

485  $SO_2+3H_2 \rightarrow H_2S+2H_2O$  R5

### 486 **5. Conclusions**

487 The effect of sulfur in reduction species on oxygen transfer capacity and heterogeneous redox 488 kinetics of a CaMn<sub>0.375</sub>Ti<sub>0.5</sub>Fe<sub>0.125</sub>O<sub>3-δ</sub> perovskite oxygen carrier was studied by using the micro-489 fluidized bed thermogravimetric analysis (MFB-TGA) technology. The results show that the sulfur 490 can enhance the oxygen transfer capacity to 5 wt% from the original 4 wt %, but impede both the 491 oxidization and reduction rates for the temperatures lower than 1173 K. When the temperature is 492 higher than 1173 K, the sulfur has almost no effect on the H<sub>2</sub> reduction kinetics, but the oxidation 493 kinetics decreases by 50% as compared to that of without sulfur. Nevertheless, the oxidation for the 494 poisoned OCM is still fast enough and can achieve 4 wt% oxygen transfer capacity within 8 seconds. 495 Once removing the SO<sub>2</sub> in the reducing gas, the oxygen transfer capacity and chemical performance 496 are capable of being completely restored. The CaMn<sub>0.375</sub>Ti<sub>0.5</sub>Fe<sub>0.125</sub>O<sub>3- $\delta$ </sub> perovskite oxygen carrier 497 possesses an interestingly chemical memorable ability under alternative reduction condition with 498 and without SO<sub>2</sub>. There is a significant effect of the reaction temperature on the critical conversions 499 and reaction rates for the oxidization and reduction steps, except for the oxidization of the recovered 500 OCM. The heterogeneous reaction kinetics at different temperatures was successfully modeled by 501 the bubble bed reactor model coupled with a two-stage kinetic model, and the kinetic parameters, 502 activation energies and pre-exponential factors were obtained from modelling. It was found that the 503 reaction activation energies increase due to the effect of SO<sub>2</sub>. And elevating temperature would 504 remove SO2 poisoning effect. Combining X-ray diffraction (XRD), in situ diffuse reflectance 505 infrared Fourier transform spectroscopy (in-situ DRIFTS), and scanning electron microscopy506 energy dispersive spectrometer (SEM-EDS), we revealed that the majority of sulfur (S) formed a 507 sulfate/sulfide shell on the particle surface, whereas the minority sulfur (S) entered the particle 508 interior and further affected the lattice structure. The perovskite structure was capable of totally 509 automatic recovery from the poisoned state as well, by means of continuous redox cycles without 510 SO<sub>2</sub>.

## 511 ACKNOWLEDGMENTS

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## 516 Nomenclature

$A_{\rm chem}$	pre-exponential factor of the fast reaction stage, $m^3/mol/s$
$A_{ m diff}$	pre-exponential factor of the slow reaction stage, $m^3/mol/s$
$A_{ m L}$	pre-exponential factor of the critical conversion
$C_{\mathrm{b},i}$	concentration of gas <i>i</i> in the bubble phase, $mol/m^3$
$C_{\mathrm{e},i}$	concentration of gas <i>i</i> in the emulsion phase, mol/m <sup>3</sup>
$C_{\mathrm{eq},i}$	equilibrium concentration of gas $i$ , mol/m <sup>3</sup>
$d_{ m p}$	diameter of the oxygen carrier particle, m
$D_i$	gas <i>i</i> molecular diffusivity, $m^2/s$
$E_{\rm chem}$	activation energy of the fast reaction stage, kJ/mol
$E_{\rm diff}$	activation energy of the slow reaction stage, kJ/mol
$E_{\rm L}$	activation energy of the critical conversion, kJ/mol

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fa	volume fraction of oxygen carrier particles in all solid volume
k <sub>chem</sub>	reaction rate constant of the fast reaction stage, $m^3/mol/s$
$k_{ m diff}$	reaction rate constant of the slow reaction stage, $m^3/mol/s$
$k_{ m g}$	external mass transfer coefficient, m/s
K <sub>be</sub>	interchange coefficient between the bubble and emulsion phase, s <sup>-1</sup>
Kr	reaction rate constant of particles in the emulsion phase, s <sup>-1</sup>
$K_{ m ri}$	reaction rate constant of particles, s <sup>-1</sup>
m(t)	mass measured by the weight transducer at time $t$ , g
mo	whole mass of the reactor and oxidized oxygen carrier samples, g
m <sub>OC</sub>	mass of the added oxygen carrier samples, g
m <sub>r</sub>	whole mass of the reactor and reduced oxygen carrier, g
$M_i$	molar mass of <i>i</i> , kg/mol
Rg	ideal gases constant, J/mol/K
R <sub>OC,t</sub>	total oxygen transfer capacity, wt.%
Sh	Sherwood number
Т	temperature in Kelvin, K
$U_{0}$	superficial gas velocity, m/s
$U_{\mathfrak{b}}$	bubble rise velocity, m/s
${U}^*_{ m b}$	effective gas velocity in the bubble phase, m/s
$U_{ m mf}$	minimum fluidization velocity, m/s
Х	conversion level of oxidation or reduction reaction at time $t$
X <sub>chem</sub>	conversion level of the fast reaction stage at time t

$X_{\rm diff}$	conversion level of the slow reaction stage at time t
Xe	equilibrium conversion level of an oxygen carrier
Xo	conversion level of oxidation reaction at time <i>t</i>
Xr	conversion level of reduction reaction at time <i>t</i>
α	stoichiometric coefficient
$ ho_{ m s}$	particle density of oxygen carrier, kg/m <sup>3</sup>
${\cal E}_{ m mf}$	voidage of fluidized bed
θ	fraction of the bubble phase
Ψ	ratio of the conversion level at the fast reaction stage to the conversion level

## 518 **REFERENCES**

519 [1] L. Zeng, Z. Cheng, J. A. Fan, L. S. Fan, J. Gong, Metal oxide redox chemistry for chemical

520 looping processes, Nat. Rev. Chem. 2(11) (2018) 349-364.

- 521 [2] A. Lyngfelt, B. Leckner, T. Mattisson, A fluidized-bed combustion process with inherent CO<sub>2</sub>
- separation; application of chemical-looping combustion, Chem. Eng. Sci. 56(10) (2001) 3101-3113.
- 523 [3] J. Adanez, A. Abad, F. Garcia-Labiano, P. Gayan, F. Luis, Progress in chemical-looping
- 524 combustion and reforming technologies, Prog. Energ. Combust. 38(2) (2012) 215-282.
- 525 [4] M. Tang, L. Xu, M. Fan, Progress in oxygen carrier development of methane-based chemical-
- 526 looping reforming: A review, Appl. Energy 151 (2015) 143-156.
- 527 [5] L. Liu, Z. Li, L. Wang, Z. Zhao, Y. Li, N. Cai, MgO–Kaolin-supported manganese ores as oxygen
- 528 carriers for chemical looping combustion, Indust. Eng. Chem. Res. 59(15) (2019)7238-7246.
- 529 [6] M. Arjmand, H. Leion, T. Mattisson, A. Lyngfelt, Investigation of different manganese ores as

- 530 oxygen carriers in chemical-looping combustion (CLC) for solid fuels, Appl. Energy 113 (2014)
  531 1883-1894.
- 532 [7] M. Luo, S. Wang, L. Wang, M. Lv, Reduction kinetics of iron-based oxygen carriers using
- 533 methane for chemical-looping combustion. J. Power Sources 270 (2014) 434-440.
- 534 [8] C. R. Forero, P. Gayán, F. García-Labiano, L. F. De Diego, A. Abad, J. Adánez, Effect of gas
- 535 composition in chemical-looping combustion with copper-based oxygen carriers: fate of sulphur,
- 536 Int. J. Greenh. Gas Con. 4(5) (2010) 762-770.
- 537 [9] P. Gayán, F. Luis, F. García-Labiano, J. Adánez, A. Abad, C. Dueso, Effect of support on
- reactivity and selectivity of Ni-based oxygen carriers for chemical-looping combustion, Fuel 87(12)
- 539 (2008) 2641-2650.
- 540 [10] R. Siriwardane, H. Tian, T. Simonyi, J. Poston, Synergetic effects of mixed copper-iron oxides
- 541 oxygen carriers in chemical looping combustion, Fuel 108(2013) 319-333.
- 542 [11] V. Frick, M. Rydén, H. Leion, Investigation of Cu-Fe and Mn-Ni oxides as oxygen carriers for
- 543 chemical-looping combustion, Fuel Process. Technol. 150 (2016) 30-40.
- 544 [12] P. N. Mohammad, H. Leion, M. Rydén, T. Mattisson, Combined Cu/Mn oxides as an oxygen
- 545 carrier in chemical looping with oxygen uncoupling (CLOU), Energy fuels 27(10) (2013) 6031-
- 546 6039.
- 547 [13] X. Zhu, K. Li, L. Neal, F. Li, Perovskites as geo-inspired oxygen storage materials for chemical
- 548 looping and three-way catalysis: a perspective, Acs Catal. 8(9) (2018) 8213-8236.
- 549 [14] K. R. Poeppelmeier, M. E. Leonowicz, J. C. Scanlon, J. M. Longo, W. B. Yelon, Structure
- determination of CaMnO<sub>3</sub> and CaMnO<sub>2.5</sub> by X-ray and neutron methods. J. Solid State Chem. 45(1)
- 551 (1982) 71-79.

- 552 [15] R. Ruhl, J. Song, V. Thoréton, S. P. Singh, K. Wiik, Y. Larring, H. J. Bouwmeester, Structure,
- 553 electrical conductivity and oxygen transport properties of perovskite-type oxides CaMn<sub>1-x-y</sub>Ti<sub>x</sub>
- 554 Fe<sub>y</sub>O<sub>3-δ</sub>, Phys. Chem. Chem. Phys. 21(39) (2019) 21824-21835.
- 555 [16] W. Xing, M. L. Fontaine, Z. Li, J. M. Polfus, Y. Larring, C. Denonville, R. Bredesen,
- 556 Asymmetric tubular CaTi<sub>0.6</sub>Fe<sub>0.15</sub>Mn<sub>0.25</sub>O<sub>3-δ</sub> membranes: Membrane architecture and long-term
- 557 stability, J. Membrane Sci. 548 (2018) 372-379.
- 558 [17] A. Abad, A. Cabello, P. Gayán, F. García-Labiano, L. F. de Diego, T. Mendiara, J. Adánez,
- 559 Kinetics of CaMn<sub>0.775</sub>Ti<sub>0.125</sub>Mg<sub>0.1</sub>O<sub>2.9-δ</sub> perovskite prepared at industrial scale and its implication on
- the performance of chemical looping combustion of methane, Chem. Eng. J. (2020) 124863.
- 561 [18] L. Liu, Z. Li, Z. Li, Y. Larring, N. Cai, Heterogeneous reaction kinetics of a perovskite oxygen
- 562 carrier for chemical looping combustion coupled with oxygen uncoupling, Chem. Eng. J. 128054563 (2020).
- 564 [19] A. Abad, F. García-Labiano, P. Gayán, L. F. de Diego, J. Adánez, Redox kinetics of CaMg<sub>0</sub>.
- <sup>1</sup>Ti<sub>0.125</sub>Mn<sub>0.775</sub>O<sub>2.9-δ</sub> for chemical looping combustion (CLC) and chemical looping with oxygen
   uncoupling (CLOU), Chem. Eng. J. 269 (2015) 67-81.
- 567 [20] Y. Larring, M. Pishahang, J. Tolchard, A. M. Lind, M. F. Sunding, R. E. Stensrød, K. Albertsen,
- 568 Fabrication process parameters significantly affect the perovskite oxygen carriers materials (OCM)
- 569 performance in chemical looping with oxygen uncoupling (CLOU), J.Therm. Anal. Calorim. 140
- 570 (2019) 1-13.
- 571 [21] M. Arjmand, A. Hedayati, A.M. Azad, H. Leion, M. Rydén, T. Mattisson, Ca<sub>x</sub>La<sub>1-x</sub>Mn<sub>1-y</sub>M<sub>y</sub>O<sub>3-</sub>
- 572  $\delta$  (M = Mg, Ti, Fe, or Cu) as Oxygen Carriers for Chemical-Looping with Oxygen Uncoupling
- 573 (CLOU), Energy Fuels 27 (2013) 4097-4107.

- 574 [22] L. F. de Diego, A. Abad, A. Cabello, P. Gayán, F. García-Labiano, J. Adánez, Reduction and
- 575 oxidation kinetics of a  $CaMn_{0.9}Mg_{0.1}O_{3-\delta}$  oxygen carrier for chemical-looping combustion, Ind. Eng.
- 576 Chem. Res. 53(1) (2014)., 87-103.
- 577 [23] M. Pishahang, Y. Larring, M. Sunding, M. Jacobs, F. Snijkers, Performance of Perovskite-Type
- 578 Oxides as Oxygen-Carrier Materials for Chemical Looping Combustion in the Presence of H2S.
- 579 Energy Tech. 4(10) (2016) 1305-1316.
- 580 [24] R. F. Pachler, S. Penthor, K. Mayer, H. Hofbauer, Fate of sulfur in chemical looping combustion
- of gaseous fuels using a Perovskite oxygen carrier, Fuel 241 (2019) 432-441.
- 582 [25] A. Cabello, A. Abad, P. Gayán, L. F. de Diego, F. García-Labiano, J. Adánez, Effect of operating
- 583 conditions and  $H_2S$  presence on the performance of  $CaMg_{0.1}Mn_{0.9}O_{3-\delta}$  perovskite material in
- 584 chemical looping combustion (CLC), Energy fuels 28(2) (2014) 1262-1274.
- 585 [26] L. Liu, Z. Li, Z. Li, Y. Larring, Y. Li, N. Cai, Fast redox kinetics of a perovskite oxygen carrier
- 586 measured using micro-fluidized bed thermogravimetric analysis, Proc. Combust. Inst. (2020).
- 587 [27] Y. Li, H. Wang, W. Li, Z. Li, N. Cai, CO<sub>2</sub> Gasification of a Lignite Char in Microfluidized Bed
- 588 Thermogravimetric Analysis for Chemical Looping Combustion and Chemical Looping with
- 589 Oxygen Uncoupling, Energy fuels 33(1) (2018) 449-459.
- 590 [28] Y. Li, Z. Li, L. Liu, N. Cai, Measuring the fast oxidation kinetics of a manganese oxygen carrier
- using microfluidized bed thermogravimetric analysis, Chem. Eng. J. 385 (2020) 123970.
- 592 [29] D. Kunii, O. Levenspiel, Fluidization Engineering, Butterworth-Heinemann, Newton, U.S.A.
- 593 1991, p. 155.
- 594 [30] Z. Li, General rate equation theory for gas-solid reaction kinetics and its application to CaO
- 595 carbonation, Chem. Eng. Sci. 227(2020) 115902.

- 596 [31] D. K. Lee, An apparent kinetic model for the carbonation of calcium oxide by carbon dioxide,
- 597 Chem. Eng. J. 100(1-3) (2004) 71-77.
- 598 [32] M. Ishida, C. Y. Wen, Comparison of zone-reaction model and unreacted-core shrinking model
- in solid—gas reactions—I isothermal analysis, Chem. Eng. Sci. 26(7) (1971) 1031-1041.
- 600 [33] S. K. Bhatia, D. D. Perlmutter, A random pore model for fluid-solid reactions: I. Isothermal,
- 601 kinetic control, AIChE J. 26(3) (1980) 379-386.
- 602 [34] J. C. Shelley, M. Y. Shelley, R. C. Reeder, S. Bandyopadhyay, M. L. Klein, A coarse grain
- model for phospholipid simulations, J. Phys. Chem. B 105(19) (2001) 4464-4470.
- 604 [35] J. Madarász, T. Leskelä, J. Rautanen, L. Niinistö, Oxidation of alkaline-earth-metal sulfide
- 605 powders and thin films, J. Mater. Chem. 6(5) (1996) 781-787.
- 606 [36] M. Zheng, S. Zhong, K. Li, H. Wang, H. Liu, Y. Wei, X. Zhu, Characteristics of CaS-CaO
- 607 Oxidation for Chemical Looping Combustion with a CaSO<sub>4</sub>-Based Oxygen Carrier, Energy Fuels
- 608 31(12) (2017) 13842-13851.